ARTICLE Stable High-Energy Density Super-Atom Clusters of Aluminum Hydride

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(Dated: Received on November 8, 2011; Accepted on February 12, 2012)

With the concept of super-atom, first principles calculations propose a new type of super stable cage clusters Al_nH_{3n} that are much more energetic stable than the well established clusters, Al_nH_{n+2} . In the new clusters, the aluminum core-frame acts as a super-atom with nvertexes and 2n Al-Al edges, which allow to adsorb n hydrogen atoms at the top-site and 2nat the bridge-site. Using $Al_{12}H_{36}$ as the basic unit, stable chain structures, $(Al_{12}H_{36})_m$, have been constructed following the same connection mechanism as for $(AlH_3)_n$ linear polymeric structures. Apart from high hydrogen percentage per molecule, calculations have shown that these new clusters possess large heat of formation values and their combustion heat is about 4.8 times of the methane, making them a promising high energy density material.

Key words: High-energy density, Super-atom, First principles, Cluster of aluminum hydride

I. INTRODUCTION

Aluminum and aluminoborane clusters with high hydrogen capacity and attractive hydrogen desorption temperature are considered to be good candidates for hydrogen storage [1]. Aluminum clusters can also be used as catalysis and propulsion material [2, 3]. The central focus on many theoretical and experimental studies is to find new stable clusters, to understand the formation mechanism and to discover their unique properties. For aluminum hydride clusters of rich Al phase $(1 \le m \ll n \text{ in } Al_n H_m)$, the free electron is dominant in the system and the stability can be well explained by the jellium model [4, 5]. With more H atoms adsorbed on the aluminum substrate, electrons in the hydrogenated system are believed to be more localized. However, the mechanism for the stability of rich H phase (n < m in Al_nH_m) is still largely unknown. The most exciting findings so far is probably the experimental observation of an unexpected stable specie, Al_4H_6 [3], which was found to obey the so-called Wade-Mingos rules [6-8]. Following this rule, it was confirmed that in general Al_nH_{n+2} clusters should be stable [9, 10]. However, one can notice that in each Al_nH_{n+2} cluster, there are still (2n-2) free electrons left, which can in principle be saturated by extra hydrogen atoms if we consider the Al cluster as a super-atom.

In this work, we show new stable clusters for alu-

minum hydrides that fulfill the super-atom principle. The structures of two stable cage aluminum hydride Al_6H_{18} and $Al_{12}H_{36}$ have been analyzed in detail. We have also shown that the cage $Al_{12}H_{36}$ cluster can be used as a basic unit to form $(Al_{12}H_{36})_m$ chain structure with the same bonding mechanism as that of $(AlH_3)_n$ polymers. The combustion heat of new clusters has also been estimated and its potential usage as the energy material is discussed.

II. COMPUTATION DETAILS

Structure optimization and electronic structure calculations were performed using Perdew-Wang exchange and gradient-corrected correlation functional (PW91) [11] with 6-311G(d,p) Gaussian basis set as implemented in Gaussian 03 package [12]. The PW91 functional is known to give good description for the geometry and electronic structure of aluminum hydrides [9, 13]. The Geometry optimization has been done without symmetry constraint. Moreover, analytic Hessian matrix has also been calculated to guarantee that the obtained final structures are true minimum.

III. RESULTS AND DISCUSSION

A. Stable structures of Al₆H₈ and Al₁₂H₃₆

After considering many different structural arrangements, we finally find the core structure of the Al_n cluster that is capable of adsorbing 3n hydrogen atoms. Two best examples are Al_6H_{18} and $Al_{12}H_{36}$ as shown in

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FIG. 1 The optimized structures of Al_6H_{18} and $Al_{12}H_{36}$, and the corresponding infrared spectra.

Fig.1. In these structures, one can see n top-site hydrogen atoms and 2n bridge hydrogen atoms. For Al₆H₁₈, the optimized geometry is of octahedral symmetry. The Al-Al bond length among Al_6H_{18} is 3.19 Å, much longer than that in pure aluminum clusters, demonstrating that the direct interaction between Al atoms is greatly weakened due to the absorption of abundant hydrogen atoms. Actually, with hydrogen atoms adsorbed on bridge sites, the original Al–Al linkage is replaced by three centers Al-H-Al bridge bond, as what was found in Al_4H_6 [3]. The terminal Al-H bond length is 1.59 Å, which is a typical covalence bond length between Al and H atoms. The bond length between bridge H and Al atoms among Al_6H_{18} is 1.75 Å. Its infrared spectrum shows two main peaks at around 900 and 1800 cm^{-1} , resulting from the motion of hydrogen atoms in the system. For $Al_{12}H_{36}$, the most stable cage structure has a C_2 symmetry, as shown in Fig.1. Although an octahedral symmetric configuration with 12 terminal and 24 bridge H atoms was constructed as an initial configuration, the geometry relaxation has led to the reduced symmetry. The predicted infrared spectrum for this stable $Al_{12}H_{36}$ cluster is broadened because of its low symmetry, compared with Al₆H₁₈. The average Al–Al bond length among $Al_{12}H_{36}$ is 3.47 Å, which is longer than that in Al_6H_{18} , inferring that with the cluster size increasing, the interaction between two adjacent aluminum atoms gets even weaker. The average bond length between bridge H and Al atoms among $Al_{12}H_{36}$ is about 1.76 Å and the terminal Al-H bond length remains 1.59 Å. From these geometry parame-



FIG. 2 The optimized structures of Al₆H₆, Al₆H₈, Al₆H₁₀, Al_6H_{12} , Al_6H_{16} , and Al_6H_{18} .

ters, one can conclude that these two cage clusters have the same structure characteristics.

B. Al_6H_m and $Al_{12}H_m$ series

The discovery of the stable Al₄H₆ cluster has attracted considerable attention [3]. Such an unexpected cluster was found to follow the so-called Wade-Mingos rule [6]. For rich H phase, the Wade-Mingos clusters, Al_nH_{n+2} , have become the most important species in the whole aluminum hydride family [9]. It is worth noting that the Al_6H_{18} super-atom cluster possesses the same octahedral Al skeleton as that of Wade-Mingos cluster, Al_6H_8 , whereas the Al frames for $Al_{12}H_{36}$ and $Al_{12}H_{14}$ are different. The Wade-Mingos cluster Al₁₂H₁₄ has an icosahedral Al frame, while it is an octahedral frame for super-atom cluster $Al_{12}H_{36}$. In order to understand the formation of the new super-atom clusters, we have examined the structural evolution of the Al_6H_m and $Al_{12}H_m$ series. With the geometry optimization, for the Al_6H_m series, stable structures with m=6, 8, 10, 12, 16, and 18, have been obtained, whilefor the $Al_{12}H_m$, structures of m=12, 14, 16, 20, and 36,are identified.

We start with Al_6H_6 cluster for the Al_6H_m series as shown in Fig.2. It is found that the optimized Al_6H_6 is of D_{3d} symmetry and each hydrogen atom interacts with aluminum substrate through covalence bond on each top site. The two added hydrogen atoms in Al_6H_8 occupy the hollow sites in two opposite faces and the system of the whole cluster remains the same as that of Al_6H_6 . For Al_6H_{10} cluster, the added four new hydrogen atoms prefer to the bridge sites and the Al corestructure is slightly distorted, resulting in a C_{2h} symmetry. With two more H atoms supplied, the Al skeleton of the Al_6H_{12} cluster undergoes a noticeable change. The triangle connection of the Al atoms turns into the quadrilateral and the total symmetry of the Al_6H_{12} be-

Ke-Yan Lian et al.



FIG. 3 The optimized structures of $Al_{12}H_{12}$, $Al_{12}H_{14}$, $Al_{12}H_{16}$, and $Al_{12}H_{20}$.

comes C_{2v} . With many attempts, we failed to find stable 3D structure for Al_6H_{14} , for which the aluminum core-frame is dissolved after a long time optimization. For Al_6H_{16} , the Al skeleton returns back to its original appearance as in Al_6H_6 . The extra hydrogen atoms are all adsorbed at the bridge sites. The geometry of superatom cluster Al_6H_{18} resembles well the structure of the cluster Al_6H_{16} with D_{2h} symmetry.

The related structures of $Al_{12}H_m$ clusters are collected in Fig.3. Our optimized Al₁₂H₁₂ cluster possesses a D_{5d} symmetry, in close agreement with the earlier reported structure of I_h symmetry [14]. In this case, each hydrogen atom is bonded to an aluminum atom on the top site. The two extra hydrogen atoms in the optimized $Al_{12}H_{14}$ of D_{3d} symmetry occupy two opposite hollow sites, similar to the case of Al₆H₈. However, for Al₁₂H₁₆ cluster all four extra hydrogen atoms prefer to the bridge sites with approximately icosahedral skeleton. When four more hydrogen atoms are added to form Al₁₂H₂₀, the symmetry of Al₁₂ core-frame is gradually changed from semi-icosahedral to semi-octahedral to provide suitable edges for adsorbing hydrogen atoms. We have found that further adding hydrogen atoms between 20 and 36, the cluster does not lead to any stable structures and the aluminum skeleton is always broken down. The $Al_{12}H_{36}$ is a stable structure with octahedral aluminum core-frame. In general, for Al_{12} cluster, the octahedral core-frame tends to adsorb more hydrogen atoms than the icosahedral one. It is also important to note that not all Al_nH_m cluster has a stable configuration. There are certain magic numbers involved.

There are several important parameters that are useful for judging the stability of a cluster, namely the energy gap (E_g), the vertical electron affinity (VEA) and the vertical ionization potential (VIP). Normally, a cluster with larger E_g , lower VEA and higher VIP is

TABLE I Calculated HOMO-LUMO energy gap (E_g) , BE, VEA, and VIP for different Al₆H_m and Al₁₂H_m clusters. All values are in eV.

	$E_{\rm g}$	BE	VEA	VIP
Al_6H_6	0.54	2.33	2.86	7.43
$\mathrm{Al}_{6}\mathrm{H}_{8}$	2.83	2.43	0.73	7.51
$\mathrm{Al}_{6}\mathrm{H}_{10}$	2.08	2.39	1.50	7.62
$\mathrm{Al}_{6}\mathrm{H}_{12}$	3.23	2.47	0.79	7.97
$\mathrm{Al}_{6}\mathrm{H}_{16}$	1.44	2.39	2.12	7.62
$\mathrm{Al}_{6}\mathrm{H}_{18}$	3.28	2.44	1.85	9.14
$Al_{12}H_{12} \\$	0.23	2.52	4.21	7.91
$\mathrm{Al}_{12}\mathrm{H}_{14}$	2.50	2.56	1.91	7.84
$\mathrm{Al}_{12}\mathrm{H}_{16}$	2.34	2.50	2.05	7.75
$Al_{12}H_{20} \\$	1.77	2.43	2.85	7.93
$\mathrm{Al}_{12}\mathrm{H}_{36}$	4.08	2.46	1.41	9.00

often more stable. The calculated values for Al_6H_m and $Al_{12}H_m$ series are summarized in Table I. The Wade-Mingos clusters Al_6H_8 and $Al_{12}H_{14}$ are known to be very stable, which are clearly demonstrated by their energetic properties. By using Al_6H_8 as the reference one, we can immediately see that Al_6H_6 , Al_6H_{10} and Al_6H_{16} are less stable species, while Al_6H_{12} and Al_6H_{18} are comparably stable, if not more stable. For the $Al_{12}H_m$ series, it can be found that the Wade-Mingos cluster $Al_{12}H_{14}$ is much more stable than $Al_{12}H_{12}$ and $Al_{12}H_{20}$, slightly more stable than $Al_{12}H_{16}$. However, it is definitely much less stable than $Al_{12}H_{36}$. In general, both super-atom clusters, Al_6H_{18} and $Al_{12}H_{36}$, possess very good energetic properties that guarantees the stability. It is worth noting that the large gap energies of 3.28 and 4.06 eV for super-atom clusters Al_6H_{18} and $Al_{12}H_{36}$ are comparative with that of $(AlH_3)_n$ polymer. Based on all these results, one can at least conclude that it is possible to find more stable hydrogen rich aluminum hydrides than what the Wade-Mingos rule allows.

C. Chemical bond between the hydrogen and the aluminum atoms

We have used different ways to illustrate the chemical bond between the hydrogen and the aluminum atoms. The first one is the deformation of electron density (ED), which is obtained by a simple relation, $ED_{Al_nH_{3n}}-ED_{Al_nH_n}-ED_{2nH}$, *i.e.* subtracting electron density of Al_nH_n (only terminal Al-H bonds involved) and those individual bridge hydrogen atoms from the total electron density of Al_nH_{3n} . With it, one can determine how these bridge hydrogen atoms are involved in its chemical bonding with the aluminum substrate, and verify whether the valence bond between top site hydrogen atoms and aluminum atoms are affected by bridge hydrogen atoms.

In Fig.4, we present both depletion and excess of elec-



FIG. 4 (a) The depletion and (b) excess of electron density with the bridged H atoms adsorbed on $Al_{12}H_{36}$.

tron density for $Al_{12}H_{36}$ as a result of adsorbed bridge H atoms. We can see that there is no obvious electron deformation between the terminal Al-H bonds. The electron density around the hollow site of the aluminum substrate decreases, but it increases around the H atoms on the bridges. That is to say, the added H atoms on bridges do not affect the covalent bond between the terminal Al and H but completely mix with Al atoms through the covalent bonding and contribute to the formation of three centers Al-H-Al. Accompanying with 2n bridged hydrogen atoms, all free electrons in aluminum clusters are totally localized around three centers Al-H-Al bridge bonds.

We have also plotted out in Fig.5 the orbital distributions of Al₁₂H₁₂, Al₁₂H₁₄, and Al₁₂H₃₆ to further examine the nature of the chemical bonds in these clusters. In Fig.5 each orbital is partitioned into three parts that are contributed from the aluminum core-frame, top-site hydrogen atoms and bridge/hollow-site hydrogen atoms, respectively. It can be seen that the energy positions of hydrogen-induced orbitals are highly dependent on the adsorption sites of hydrogen atoms. The orbitals involved of terminal Al-H bonds are confined to the energy region from -11 to -7 eV in all these clusters, while the contribution of bridge hydrogen atoms spread over the whole energy spectrum, well mixed with those oribitals of Al atoms. This is particularly obvious in Al₁₂H₃₆ cluster, indicating that these two types of hydrogen atoms interact with the aluminum substrate differently. The hydrogen atoms on the top site form covalence bonds with aluminum atoms and these covalence bonds are not affected by the presence of bridged hydrogen atoms. The bridged hydrogen atoms contribute their electrons to the whole system. In $Al_{12}H_{36}$, the contributions of hydrogen atoms in all oribitals are almost equal to those of aluminum atoms.

It can be seen from the energy level distribution of $Al_{12}H_{12}$ in Fig.5, the energy gap between the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO) is very small, but the gap between LUMO and (LUMO+1) gap quite large. This indicates that the $Al_{12}H_{12}$ is not stable in general. To add two hydrogen atoms, the LUMO of $Al_{12}H_{12}$ is filled up and the resulting $Al_{12}H_{14}$ cluster become very stable. This picture is consistent with the pro-



FIG. 5 The energy level distribution with partitioned contributions from the aluminum substrate (Al), top-site hydrogen atoms (H_{top}) and bridge/hollow-site hydrogen atoms (H_{brg}) for Al₁₂H₁₂, Al₁₂H₁₄, and Al₁₂H₃₆, respectively.

posed Wade-Mingos rules for Al_4H_6 [3]. Wade-Mingos rules, also known as polyhedral skeletal electron pair approach, are empirical theory based on bonding models. With accurate molecular orbital calculation, we can now provide a clear molecular orbital view on the stability of $Al_{12}H_{14}$.

With the concept of super-atom, we can also relate the stability of a cluster to its special shell structure of electronic distribution. In Fig.6, we have plotted out the orbital pictures of all valence orbitals for $Al_{12}H_{12}$ and $Al_{12}H_{14}$ and calibrated them with corresponding atomic orbitals. It can be seen that the ground state electronic structure of $Al_{12}H_{12}$ can be presented by a super-atom with a configuration of $1s^21p^61d^{10}1f^62s^22p^62d^{10}1f^6$. It is known that the filled f shell should hold 14 electrons, while there exist only 12 f electrons in $Al_{12}H_{12}$. Moreover, the LUMO of $Al_{12}H_{12}$ is also f orbital and LUMO+1 is of s character. The small HOMO-LUMO gap and large LUMO-LUMO+1 gap could be understood from the fact that they fall into either the same or different shells. When two hydrogen atoms are added, the symmetry of the system is reduced from D_{5d} to D_{3d} , resulting in the energy adjustment for related molecular orbital. The ground state electronic structure of $Al_{12}H_{14}$ has a configuration of $1s^21p^61d^{10}1f^62s^22p^22d^22p^41f^22d^81f^6$, consisting of fully occupied two s shells, two p shell, one d shell and one f shell. We can say that the extremely stability of Al₁₂H₁₄ comes from its closed shell-like electronic distribution.



FIG. 6 The molecular orbital pictures of valence orbitals for $Al_{12}H_{12}$ and $Al_{12}H_{14}$.

D. Thermodynamic properties of Al₆H₈ and Al₁₂H₃₆

One important thermodynamic property of a molecule is the heat of formation. We have used atomization reaction method to calculate it. In this approach, the Al_nH_m is decomposed into Al and H atoms, and the heat of formation for Al_nH_m is calculated with the help of the reaction: $Al_nH_m \rightarrow nAl+mH$. We firstly calculate the heat of formation for molecular Al_nH_m at 0 K ($\Delta_f H_M^{\oplus}(0 \text{ K})$), following the relationship:

$$\Delta_{\rm f} H_{\rm M}^{\rm e}(0 \,\mathrm{K}) = \sum x \Delta_{\rm f} H_{\rm X}^{\rm e}(0 \,\mathrm{K}) - D_0(\mathrm{M}) \tag{1}$$

here, $\Delta_{\rm f} H^{\diamond}_{\rm X}(0 \text{ K})$ stands for the heat of formations for constituent atomic elements (X) at 0 K which are taken from the reference [15]; $D_{0,{\rm Al}_n{\rm H}_m}$ stands for the atomization energy which is calculated by the corresponding energy difference from the inclusion of zeor-point vibration energy correction, *i.e.*,

$$D_{0,\mathrm{Al}_{n}\mathrm{H}_{m}} = (nE_{0,\mathrm{Al}} + mE_{0,\mathrm{H}}) - (E_{0,\mathrm{Al}_{n}\mathrm{H}_{m}} + E_{\mathrm{ZPE},\mathrm{Al}_{n}\mathrm{H}_{m}})$$
(2)

We introduce temperature correction represented as $H^{\bullet}_{\text{Al}_{n}\text{H}_{m}}(298 \text{ K}) - H^{\bullet}_{\text{Al}_{n}\text{H}_{m}}(0 \text{ K})$ and $\sum x[H^{\bullet}_{X}(298 \text{ K}) - H^{\bullet}_{X}(298 \text{ K})]$

DOI:10.1088/1674-0068/25/02/147-152



FIG. 7 Optimized chain structures of $(\mathrm{Al}_{12}\mathrm{H}_{36})_2$ and $(\mathrm{Al}_{12}\mathrm{H}_{36})_3.$

 $H^{\circ}_{\rm X}(0 \text{ K})]$, here, the first part is the enthalpy correction for the molecule, obtained from the frequency analysis; the latter part is the enthalpy corrections for the aluminum and hydrogen atoms, referring to the chemical database [16]. Finally, we get the the heat of formation for molecular Al_nH_m at 298 K from the relationship:

$$\Delta_{\rm f} H^{\rm e}(298 \text{ K}) = \Delta_{\rm f} H^{\rm e}_{{\rm Al}_n {\rm H}_m}(0 \text{ K}) + [H^{\rm e}_{{\rm Al}_n {\rm H}_m}(298 \text{ K}) - H^{\rm e}_{{\rm Al}_n {\rm H}_m}(0 \text{ K})] - \sum_{\rm X} x [H^{\rm e}_{\rm X}(298 \text{ K}) - H^{\rm e}_{\rm X}(0 \text{ K})]$$
(3)

With all these values in hand, we have obtained the heat of formation for Al_6H_{18} and $Al_{12}H_{36}$ at 298 K, which is 0.55 and 1.00 MJ/mol, respectively. These are comparable high positive heat of formation values, indicating that these clusters can be potential high energy density material. We have also calculated their combustion heat to the products, Al_2O_3 and H_2O , which are found to be 4.90 and 5.74 MJ/mol for Al_6H_{18} and $Al_{12}H_{36}$, respectively. These values are about 4.8 times of the methane, and three times of high energy material Al_4H_6 [3].

E. The formation of $(AI_{12}H_{36})_n$ chains

The crystal structures of aluminum trihydride possess several different polymorphs with covalent AlH_3 cluster as the smallest structure unit. We have found that the super-atom clusters, $Al_{6n}H_{18n}$, can function the same way as AlH_3 to form chain-like polymeric structures with retained structure characteristics. With the number of hydrogen atoms three times as many as aluminum atoms per molecule, materials made from $Al_{6n}H_{18n}$ super-atom clusters can store up to 10.1% of hydrogen just like aluminum trihydride. The unique thermodynamic properties could make them promising high-energy high-density material (HEDM).

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We have constructed different $(Al_{12}H_{36})_n$ conformations with the $Al_{12}H_{36}$ cluster as the basic constructive unit. In Fig.7, the optimized most stable structures for $(Al_{12}H_{36})_2$ and $(Al_{12}H_{36})_3$ are illustrated. In these oligomers, two top-site hydrogen atoms on two different $Al_{12}H_{36}$ clusters have served as the connecters. The distance between the hydrogen and aluminum atoms of two clusters lies in 1.70-1.90 Å. It is important to note that for each $Al_{12}H_{36}$ molecule, apart from one top hydrogen atom involved in the connection, the entire $Al_{12}H_{36}$ cage structure remains its unique geometry characteristics. All twelve top hydrogen atoms in $Al_{12}H_{36}$ cluster can have the possibility to connect to another cluster. It is thus highly feasible to form 3-dimensional interconnected $Al_{12}H_{36}$ structures.

IV. CONCLUSION

A new type of stable cluster Al_nH_{3n} that obeys the concept of super-atom is predicted from first principles calculations. It could be another important aluminum hydride family for rich hydrogen phase apart from the well-known Wade-Mingos clusters Al_nH_{n+2} . A simple structural consideration for such new clusters is revealed, which requires an aluminum core-frame that can provide n vertexes and 2n Al–Al edges to bond hydrogen atoms. Detailed electronic structures of these new clusters have been given and illustrated with interesting atomic orbital pictures. It is found that these new clusters can form stable polymeric chain structures without causing any structural modifications. With high hydrogen percentage per molecular and very high combustion heat, these stable hydrogen rich clusters could be useful for energy applications.

V. ACKNOWLEDGEMENTS

This work was supported by the Swedish Research Council, Swedish National Infrastructure for Computing, the National Natural Science Foundation of China (No.10534010 and No.20925311), the Fundamental Research Funds for the Central Universities (No.201103255), and the China Scholarship Council.

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